Plasmon-induced efficient hot carrier generation in graphene on gold ultrathin film with periodic array of holes: Ultrafast pump-probe spectroscopy

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ABSTRACT
Using ultrafast pump-probe reflectivity with a 3.1 eV pump and coherent white light probe (1.1–2.6 eV), we show that graphene on gold nanostructures exhibits a strong coupling to the plasmonic resonances of the ordered lattice hole array, thus injecting a high density of hot carriers in graphene through plasmons. The system being studied is single-layer graphene on an ultrathin film of gold with periodic arrangements of holes showing anomalous transmission. A comparison is made with gold film with and without hole array. By selectively probing transient carrier dynamics in the spectral regions corresponding to plasmonic resonances, we show efficient plasmon induced hot carrier generation in graphene. We also show that due to high electromagnetic field intensities at the edge of the submicron holes, fast decay time (10–100 fs), and short decay length (1 nm) of plasmons, a highly confined density of hot carriers (very close to the edge of the holes) is generated by Landau damping of plasmons within the holey gold film. A contribution to transient decay dynamics due to the diffusion of the initial nonuniform distribution of hot carriers away from the hole edges is observed. Our results are important for future applications of novel hot carrier device concepts where hot carriers with tunable energy can be generated in different graphene regions connected seamlessly.

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I. INTRODUCTION
Surface plasmon polaritons (SPPs) due to their inherent property of nanoscale confinement and localization, smaller than the interacting light wavelength, show strong light-matter interaction. This leads plasmonics to a wide variety of applications from efficient harvesting of solar energy to nanoscale optical devices.1–4 Efficient generation and transfer of hot electron-hole pairs have been at the heart of efficient conversion of solar energy in photovoltaic and photocatalytic devices.5 Plasmons in nanostructures decay through Landau damping (with damping time of few tens of femtoseconds6) to generate energetic electrons, with energies larger than those of the thermal excitation (known as the hot electrons) at ambient temperatures.9–14 Hot carriers can be accessed through plasmon induced hot-electron charge transfer into adjacent interface materials.15,16 If a high conductance pathway is available, the plasmon generated hot electrons can rapidly conduct over long distances, which might also discourage the electrons once transferred from returning to the plasmonic material. In graphene, electrons move with high velocities \( v_F = 10^6 \text{ m/s} \) and behave like massless fermions, making it the most desirable material for the high-speed electronics and optoelectronics applications. A substantial challenge to the performance of most of
the graphene-based optoelectronic devices comes from the weak absorption of light (only 2.3% at normal incidence) in graphene. Thus, efficient hot electron injection in graphene through plasmons gains valuable importance. Graphene in a hybrid plasmonic structure is a promising element for high-speed electrically controllable optical and optoelectronic devices. Two dominant mechanisms to induce hot carriers in graphene are (i) direct hot electron transfer by Landau damping of plasmons in Au nanostructures and graphene covered gold nanorods, and (ii) generation of carriers by the near field enhanced electric field, as reported in graphene covered nanodisks. Ultrafast pump-probe experiments have played an important role in identifying the hot carrier transfer mechanism and dynamics.

Ever since the phenomenon of extraordinary optical transmission (EOT) was demonstrated, plasmonic structures with a periodic array of holes in metal films owing to high refractive index sensitivity (RIS) and tunable plasmonic properties have found wide applications from optical elements to chemical and biological sensors. Graphene on plasmonic nanostructures of noble metals is the simplest and effective hybrid structure for seamless integration of graphene with plasmons. Recent experiments have shown a strong light-graphene interaction enhanced by plasmons of an array of holes and nanovoids in gold. Of particular interest are the nanovoids with protruding edges which show strong graphene-plasmon interaction due to a relatively large interaction area between graphene and the strong electric field of plasmons at the voids or rims. Despite having the possibility of more strong interaction of graphene on 3D hole arrays due to large interaction area (similar to the case of nanovoids) and enhanced field intensities (because of matched plasmon energy between the two sides), this has not been explored. Most of the studies of graphene-covered periodic hole/nanovoid arrays so far have focused on the enhanced environmental sensitivity of plasmonic resonances. The ultrafast interplay of the dynamics of carriers and plasmons in the graphene-covered metal hole array is completely unexplored.

Here, we show that a high density of hot-electrons can be generated in graphene through a strong interaction in a hybrid plasmonic structure of graphene with a 3D gold hole array. Using ultrafast pump pulses (~80 fs) of energy 3.1 eV, we photoexcite carriers in the graphene-gold hole array hybrid system and probe their relaxation dynamics with coherent white light in the spectral window of 1.1 eV–2.6 eV. Notably, pump-induced reflectivity shows significant signatures in the spectral window corresponding to EOT resonances of the gold hole array originating from the carrier dynamics in graphene. A comparative study of the graphene on the gold film with and without hole array confirms the highly efficient direct plasmon induced hot carrier generation in graphene.

II. EXPERIMENTAL AND SIMULATION DETAILS

The gold hole array was fabricated on a glass substrate by a colloidal lithography technique. The hole array was prepared by spin coating the solution of polystyrene spheres of 1 μm mean diameter on a glass substrate for a hexagonal close-packed colloidal monolayer. The polystyrene sphere diameter was reduced to about 480 nm with reactive ion etching using the O₂ plasma. Subsequently, the 50 nm Au with 5 nm Ti adhesion layer was deposited on the microspheres by sputtering. The removal of polystyrene spheres by ultrasonication gives holes of 490 ± 30 nm diameter in the gold film. The atomic force microscopy (AFM) and scanning electron microscopy (SEM) were used to characterize the morphology of the fabricated hole array [see Figs. 1(a) and 1(b)]. The array of holes with an
average diameter of 490 nm are arranged in a triangular lattice of 1 µm periodicity.

Single-layer graphene samples were grown on a 25 µm thick copper foil by the chemical vapor deposition technique and transferred on gold films with and without hole array by the conventional technique. The PMMA on the graphene was removed by immersing the substrate in acetone at 60 °C, followed by annealing for 3 h at 500 °C under hydrogen and argon atmospheres. For a reference, the Au film of 50 nm thickness without hole array with Ti adhesion layer was deposited on a glass substrate. We have studied four samples: (i) 50 nm gold film with hole array (AuHA), (ii) graphene covered gold hole array (G-AuHA), (iii) 50 nm gold film (Au), and (iv) graphene covered 50 nm gold film (G-Au).

Figure 1(c) shows the Raman spectrum of G-AuHA and G-Au with an excitation wavelength of 532 nm. The peak positions of G mode (1590 cm⁻¹) and 2D mode (2680 cm⁻¹) with a width of ~31 cm⁻¹) and their intensity ratio confirm the single layer graphene on G-AuHA. The D band intensity with respect to the G-band (I(D)/I(G) ~ 0.68) shows the presence of disorder in the single layer graphene. The D' band (1621 cm⁻¹) associated with the disorder activated longitudinal optical phonons near Γ-point is also seen in G-AuHA. As the work function of graphene (~4.56 eV) is lower than that of gold (~4.83 eV), graphene on the gold film gets p-doped. Furthermore, CVD grown graphene gets unintentionally p-doped due to charge transfer from atmospheric H₂/O₂ molecules. The intensity ratio I(2D)/I(G) is ~2.7 and 2.0 for graphene on AuHA and graphene on Au, respectively. From the I(2D)/I(G) ratio, we find that graphene on Au is more p-doped than graphene on AuHA with Fermi energies at ~352 meV and ~206 meV, respectively. This is commensurate with the fact that graphene is relatively more in contact with gold in the case of G-Au than G-AuHA. A enhancement of ~2 in Raman intensity of D, G, and 2D modes in G-AuHA in comparison to G-Au is due to the enhanced electric field originating from the nanostructure. The G band of G-AuHA shows an asymmetric line shape [Fig. 1(d)]. We fit the G band with the Breit-Wigner-Fano (BWF) line shape: I(ω) = I_0[1 + (ω - ω_G)/Γ]²/[1 + [(ω - ω_G)/Γ]²] (where Γ is the linewidth, ω_G is the G band frequency, and I/λ is the interaction parameter between the phonon and electronic continuum). From the fit, we obtain ω_G = 1590.0 ± 0.3 cm⁻¹, Γ = 12.1 ± 0.3 cm⁻¹, and I/λ = 0.23 ± 0.01. From the Lorentzian fit to the symmetric G band in G-Au, the values are ω_G = 1580.0 ± 0.1 cm⁻¹ and Γ = 12.0 ± 0.2 cm⁻¹. Similar values of I/λ as seen by us have been reported for metallic single-walled carbon nanotubes (SWNTs) where a lower frequency (G band) transverse optical (TO) phonon component couples to the 1D-π plasmons. Asymmetric line shapes have been observed in graphene as a manifestation of a Fano resonance originating from the interaction of G-phonons with the excitonic states formed in undoped graphene. The interaction parameter I/λ observed by us is about a factor of 3 greater than that observed in pristine graphene with Fermi energy at the charge neutrality point. We attribute the observed BWF line shape to the interaction of G phonons with the electronic Raman background of AuHA, implying strong coupling of graphene with the plasmonic nanostructure.

Transient reflectivity measurements were done by exciting the samples with 80 fs pulses of central wavelength 400 nm (3.1 eV), obtained by frequency doubling 800 nm (1.57 eV) derived from Ti:sapphire amplifier (Spitfire from Spectra Physics, Inc.) with a BBO crystal. Pump-induced reflectivity changes were monitored for Au, G-Au, AuHA, and G-AuHA with a stable white light probe (generated by focusing a small fraction of 800 nm on a sapphire crystal) in a spectral window of 1.1 eV–2.6 eV. Pump beam with a fluence of 138 µJ/cm² (for Au, AuHA, and G-AuHA) and 201 µJ/cm² (for G-Au) with an incident angle of ~14° was used.

The steady state transmission measurements were done using a UV/VIS spectrophotometer (Lambda 900 Perkin Elmer). Optical transmission of the AuHA and G-AuHA is simulated using the 3D full wavevector FDTD method using the software package (FDTD Solutions from M/s Lumerical, Inc.). A plane polarized source with polarization along the c-axis propagating in the z-direction normal to AuHA and G-AuHA is considered. A transmission intensity monitor at the back side and a field intensity monitor at the front surface of AuHA and G-AuHA were kept to record the simulated light transmission and the electric field intensity at the surface, respectively. For accounting the irregularities in the circular shape of holes and better agreement with the experimentally observed transmission, the simulated transmission was averaged over transmission spectra from slightly elliptical holes (keeping the sum of major and minor axis radii 500 nm) with the same periodicity.

### III. RESULTS AND DISCUSSION

Figure 2(a) shows the measured transmission spectra of AuHA and G-AuHA showing extraordinary optical transmission (EOT) peaks due to the triangular lattice array of holes. For comparison, the transmission spectrum of the gold film of the same thickness is also shown. The transmission spectrum of AuHA and G-AuHA is not vertically shifted in Fig. 2(a), thus showing enhanced transmission with respect to Au. Transmission spectra of AuHA and G-AuHA show three EOT transmission peaks whose positions are related to the lattice constant a₀ of the hole array, dielectric constants of the surrounding medium (ε_d) and the metal (ε_m), and the Bragg resonance order (i, j) of the SPPs satisfying Bragg condition:

\[
\frac{\lambda}{a_0} \approx \frac{\lambda}{a_0} \approx \frac{\lambda}{a_0} = \left(\frac{4}{3}\right)^{1/2} \left(\frac{i^2 + j^2}{\epsilon_m + \epsilon_d} \right)^{1/2}.
\]

We observe three EOT peaks in the transmission spectrum at 0.6 eV, 1.0 eV, and 1.4 eV, marked as G₁, G₂, and G₃ in Fig. 2(a), respectively. The observed EOT peak positions are in good agreement with the reports on similar gold hole array films. The simulated transmission spectrum for the triangular lattice gold hole array of 500 nm diameter and 1000 nm periodicity on the glass substrate is shown in Fig. 2(b). This guides us in assigning the EOT peaks as follows (Table I): the transmission peak G₁ is due to the (1,0) AuHA/glass Bragg resonance order, the broad EOT peak G₂ is due to the overlap of (1,0)* Air/AuHA and (1,1) AuHA/glass resonance peaks, and the G₃ peak is attributed to the (2,0) AuHA/glass resonance peak. A slight blue shift of the simulated spectrum with respect to the measured one can be due to the irregularities in the long range periodicity of the triangular lattice of the hole arrays, which has not been considered in the simulation. An overall enhanced transmission of ~10% in AuHA with respect to Au is observed in the spectral window from 1.6 eV to 4.0 eV where EOT peaks are not seen. G-AuHA due to absorption...
FIG. 2. (a) Transmission spectrum of AuHA and G-AuHA showing extraordinary optical transmission (EOT) peaks due to the coupling of light with the plasmonic resonances of the ordered lattice hole array. Respective plasmonic mode indices are also indicated. For comparison, the transmission spectrum of the gold film (Au) of the same thickness is also included (the plots are not shifted vertically). (b) FDTD simulated transmission spectrum showing (1,0) AuHA/glass, (1,0)′ Air/AuHA, (1,1) AuHA/glass, and (2,0) AuHA/glass interface EOT peaks. For comparison, experimental AuHA transmission data in (a) are also included. Green shaded regions refer to the spectral window covered by the white light probe in the transient differential reflectivity measurement experiment. (c) and (d) Electric field intensity profile and plot at y = 0 for EOT (1,0)′ Air/AuHA and (1,0)′ Air/G-AuHA at 1.22 eV, respectively.

in graphene (2.3%) shows a slightly reduced transmission with respect to AuHA in the same spectral window. However, in the spectral window of 0.5 eV–1.6 eV at EOT peaks, due to the coupling of graphene with the plasmonic modes, transmission is reduced by ∼3.5%.29,30

The simulations with graphene on top of the AuHA did not show any difference in the transmission peak positions. However, the electric field intensity is enhanced due to graphene by more than ∼55%. Figures 2(c) and 2(d) show the simulated electric field intensity distribution profile on the top surface of AuHA and

<table>
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<tr>
<th>TABLE I. Electric field intensity ratio $R$ for EOT frequencies observed in the FDTD simulated transmission spectrum of AuHA and G-AuHA.</th>
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<tr>
<td>Calculated EOT peak position (eV)</td>
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<tr>
<td>0.83</td>
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<tr>
<td>1.22</td>
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<td>1.35</td>
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gets absorbed by the $5d \rightarrow 6s$ transition and excites the electron distribution out of equilibrium in the conduction band and holes in the 5d band. From the two-photon photoemission (2PPE) studies, it has been seen that the holes in the 5d state relax in a very short time, of the order of few tens of femtoseconds, via Auger processes and deliver the excess energy to the 6s band electrons. Thus, the excited 5d holes relax within the pump pulse duration. The non-Fermi electron distribution, through electron-electron interaction, redistributes its energy and attains a Fermi distribution with temperature $T_e \gg T_0$ (initial lattice temperature). The thermalization time of electrons varies from <100 fs to few hundreds of femtoseconds. The quasithermalized electron distribution then loses its energy within a few picoseconds to the lattice through electron-phonon interaction. An equilibrium with the lattice at a slightly higher temperature than $T_0$ is reached afterward followed by cooling of the hot lattice on a 100 ps time scale. Following the arguments of Sun et al., the nonthermalized distribution of carriers interacts with the phonon bath during the thermalization process. Therefore, exploring the energy exchange mechanism by considering the onset of the electron-electron and electron-phonon relaxation as two separate regimes is not accurate. However, the energy exchange mechanism can be modeled by taking two separate populations of photoexcited carriers of thermalized and nonthermalized distributions. Following Ref. 51, we fit the transient reflectivity with the following function convoluted with the Gaussian pulse:

$$I(t) = I_0 \exp \left(-\frac{(t-t_0)^2}{\sigma^2}\right)$$

with

$$\sigma = \frac{\lambda}{2\pi f_{\text{FWHM}}\cosh^{-1}(x)}$$

for the probe pulse duration $\tau$, where $\lambda$ is the wavelength of the probe pulse, $f_{\text{FWHM}}$ is the full width at half maximum of the Gaussian pulse, and $x$ is the reciprocal of the probe pulse duration. The fitting parameters are $I_0$, $t_0$, and $\sigma$. The inset in (b) and (c) shows the zoomed kinetics up to 2 ps delay at probe energies 1.3 eV and 1.45 eV, respectively.

**Figure 3(a)** shows a transient pump-induced reflectivity change as a function of probe energy in Au, AuHA, G-Au, and G-AuHA at a probe delay of 1 ps in the spectral window of 1.1–1.45 eV (NIR) and 1.7–2.8 eV (UV-VIS). The observed pump induced reflectivity change matches qualitatively in all four samples. The shape of the reflectivity change as a function of probe energy depends on the different regions of electronic density of states (DOS) interrogated by the probe pulse [as illustrated in Fig. 4(d)].

---

**Table I**

| Sample     | $R = |E|^2_{\text{edge}}/|E|^2_{\text{mid}}$ | $|E|^2_{\text{edge}}$ | $|E|^2_{\text{mid}}$ |
|------------|-------------------------------------------|------------------------|----------------------|
| Au         | 1.2                                      | 1.0                    | 1.0                  |
| AuHA       | 1.5                                      | 1.2                    | 1.2                  |
| G-Au       | 1.8                                      | 1.5                    | 1.5                  |
| G-AuHA     | 2.0                                      | 1.8                    | 1.8                  |

---

**Figure 3**

(a) 400 nm (3.1 eV) pump induced reflection change of Au, AuHA, G-AuHA, and G-Au at 1 ps probe delay. Kinetics of Au, AuHA, G-AuHA, and G-Au at probe energies (b) 1.3 eV and (c) 1.45 eV (UV-VIS). Observed pump induced reflectivity change matches qualitatively in all four samples. The shape of the reflectivity change as a function of probe energy depends on the different regions of electronic density of states (DOS) interrogated by the probe pulse [as illustrated in Fig. 4(d)].

---

**Figure 4**

(d) Fitting components contributing to the total fit of the kinetics of G-AuHA in (b) and (c). Kinetics at probe energy 2.2 eV compared for all four samples. The plasmonic signal originating from graphene is not observed in the probe region where plasmonic resonances are not present. (g) Amplitude of the fitting components as a function of probe energy in G-AuHA.
response,
\[
\Delta R/R (\hbar \omega, t) = H(t) \left( A_{\text{NT}} e^{-t/\tau_G} + A_{\text{L}} \left( 1 - e^{-t/\tau_e} \right) e^{-t/\tau} \right) + A_G e^{-t/\tau_e} + A_L,
\]
where \( H(t) \) is a Heaviside step function. \( A_{\text{NT}} \) and \( A_{\text{L}} \) are the initial transient amplitudes corresponding to nonthermalized and thermalized electronic distributions in Au, respectively. \( A_L \) is the offset amplitude due to slow cooling of the lattice. \( \tau_e \) and \( \tau_p \) are the rise time associated with the thermalization of photoexcited carriers and the time constant for the decay of the thermalized distribution due to electron-phonon interaction, respectively. \( \tau_F = (1/\tau_e + 1/\tau_p)^{-1} \) is the decay time of nonthermalized population, taking into account simultaneous interaction of nonthermalized electrons with the phonon bath. \( A_G \) and \( \tau_G \) are the amplitude and decay time of the signal from graphene considered only in the NIR region, respectively. As the amplitude of nonthermalized distribution is shown to be significant only around Fermi energy (\( \sim 2.4 \) eV) of gold,\(^{17}\) we include the contribution of \( A_{\text{NT}, \text{G}} \) only near 2.4 eV probe energy. Except for the probe energy \( \sim 2.4 \) eV, the rise time \( \tau_F \) in all four samples is limited by the instrument response function (\( \sim 115 \) fs). The rise time \( \tau_F \) is found to be \( \sim 250 \) fs near Fermi energy of gold. The long rise time observed near Fermi energy is due to the state filling effect,\(^{21}\) which blocks the relaxation of electrons close to the Fermi surface, while high-energy electrons decay on a much shorter time scales. This is also commensurate with the Fermi liquid theory which predicts that the relaxation rate varies as \((E - E_F)^2\).\(^{22,23}\)

Figures 3(b) and 3(c) show a comparison of the kinetics of photoexcited carriers in G-AuHA with AuHA, G-Au, and Au at two representative central probe energies 1.3 and 1.45 eV in the NIR region. For better S/N ratio and improved fitting, the kinetic traces have been averaged with a window of \( AE \sim 6 \) meV. The behavior of G-AuHA kinetics in this region compels us to consider an additional exponentially decaying component \((A_G e^{-t/\tau_G})\) with positive amplitude \( A_G \) and decay time \( \tau_G \) in the fitting function [Eq. (2)]. The positive signal contributing to \( \Delta R/R \) is related to the transient carrier dynamics of plasmon induced hot carriers in graphene. The positive sign of the signal originating from graphene is in agreement with the reported ultrafast study of graphene.\(^{22}\) No such positive signal in the UV-VIS region is observed for G-AuHA [Fig. 3(f)]. It can be seen from the fits that the dynamics is well described by the function [see Figs. 3(d) and 3(e)]. The fit to the experimental data was found to be excellent over the entire probe energies. Figure 3(g) shows the amplitudes \( A_G \) as a function of probe photon energy together with \( A_{\text{NT}, \text{G}} \) and \( A_L \) in G-AuHA. The relaxation time constants obtained from the fit are shown in Figs. 4(a) and 4(b).

We see that as the probe photon energy increases from 1.2 to 1.45 eV, \( \tau_G \) decreases from 350 ± 50 fs to 200 ± 20 fs. The decay time agrees well with the characteristic decay time of carriers through electron-optical phonon (e-op) scattering in CVD grown graphene.\(^{24-25}\) The e-op scattering in graphene is mainly due to zone-boundary optical phonons, and the decay rate is given by \( \frac{\tau_F}{\tau_p} = \frac{\pi}{2} e W.\)\(^{26}\) Here, \( |e| = E - E_F \) is the electron energy with respect to Fermi energy. \( W \) is the dimensionless electron-phonon coupling constant. The characteristic decay time, its dependence on probe photon energy, and e-op coupling constant obtained experimentally confirm that the positive signal seen in the hybrid G-AuHA structure in the NIR region originates from the carrier dynamics in graphene. As the additional signal is observed only in the G-AuHA sample in the NIR region having the plasmonic resonances \( \Delta \), plasmon induced hot electrons in the graphene with \( \tau_G \) as their...
relaxation time. Schematic in Fig. 4(c) illustrates the hot carrier generation mechanism in graphene. As the pump beam is not in resonance with the EOT peaks (G1, G2, and G3), the induced hot carriers in graphene cannot be due to the near field enhanced direct photoexcitation. The generation of hot carriers in graphene is due to plasmon induced interfacial hot carrier transfer from Au into the graphene through regions in contact with the protruding edges of the nanoholes having high near field intensities.

In the discussion that follows, we examine the relaxation time of thermalized carriers in gold through electron-phonon scattering $\tau_p$ and its dependence on probe energy as shown in Fig. 4(b). We see that $\tau_p$ in the UV-VIS range shows a similar behavior for Au and AuHA. Interestingly, there is a two fold increase in $\tau_p$ in the NIR region. The presence of graphene in G-Au and G-AuHA has no effect on $\tau_p$ in gold, as expected. The spectral dependence of the different regions probed in the electronic band structure of gold is schematically shown in Fig. 4(d). We will first discuss the dependence of $\tau_p$ in the UV-VIS range. $\tau_p$ increases from 0.8 ps at 2.0 eV to 1.2 ps at 2.4 eV and then decreases as probe energy further increases beyond 2.4 eV–2.8 eV. As the crystal structure of gold has only one atom per unit cell, carrier relaxes through acoustic phonon emission. The decay time varies as $\sim \sqrt{k_B T}/E$, where $E$ is the energy of electrons, $k_B$ is the Boltzmann constant, and $T$ is the lattice temperature. The smearing of Fermi-Dirac distribution of the electrons at the elevated electron temperature $T_e$ creates holes below the Fermi energy. As a consequence of this, the forbidden interband transition $5d \rightarrow 6s$ states below the Fermi energy at initial electron temperature $T_e = T_0$ are allowed at $T_e \gg T_0$, whereas the states above the Fermi energy are blocked by the phase space filling. Therefore, the probe photons with energy less than the interband transition threshold ($E_{5d} \sim 2.4$ eV) detect holes below Fermi energy, whereas those greater than the interband transition threshold detect electrons above Fermi energy. Hence, the observed electron-phonon cooling rate is higher when probed at 2.0 eV, increases with probe photon energy up to $E_p \sim 2.4$ eV, and then decreases as probe photon energy moves away from the Fermi energy up to 2.8 eV.

In the NIR region, there are three main results: (i) an overall increase of 1 ps in $\tau_p$ of AuHA as compared to Au, (ii) the linear increase in $\tau_p$ with probe photon energy, and (iii) as mentioned before, the presence of graphene in G-Au and G-AuHA has no effect on $\tau_p$. As the probe in the NIR region corresponds to intraband transition, one would expect photobleach signal ($\Delta R/R$ with a positive sign) due to the thermalized distribution of carriers near $E_f$. However, as the observed $\Delta R/R$ signal in this region is negative, a photoinduced absorption is possible due to the transition from $6s$ band to $6p$ band. The probe in this region is excited state absorption by the thermalized carriers above $E_F$ and detects the electron with energy $E = E_{6p} - E_F$, where $E_{6p} \sim 4.32$ eV is the position of the $6p$ band and $E_F$ is the probe photon energy. Higher energy probe photons therefore detect low energy electrons above the Fermi energy. As discussed earlier, the thermalized electrons decay through acoustic phonons with decay time varying as $\sim \sqrt{k_B T_e}/E$, $\tau_p$ is expected to vary as $\sim \sqrt{k_B T_e/(E_{6p} - E_F)}$. Therefore, in this region, we see $\tau_p$ increasing with probe photon energy.

We now look at an overall increase in $\tau_p$ by 1 ps in AuHA in the NIR region. The perforation of the gold film with holes cannot change the electron-phonon coupling constant of gold, and hence, the increase in $\tau_p$ cannot be associated with change in shape or filling factor. However, the plasmonic behavior of AuHA can alter the relaxation time $\tau_p$. In the NIR region, as the plasmons decay through the Landau damping mechanism, they generate hot electron-hole pairs in AuHA. The decay time and length of Landau damped plasmons are $\sim 1$ to 100 fs and $\sim 1$ nm, respectively. With the electromagnetic field intensity due to plasmons being high at the hole edges [Figs. 2(c) and 2(d)], an initial population of hot electrons confined very close to the edge of the holes is generated from the plasmon decay within the pulse duration. This gives rise to a nonuniform distribution of hot carriers between the holes. The hot carriers diffuse from their high density regions to form spatially uniform distribution of hot electrons. The characteristic hot carrier diffusion time is given by $\tau_d = C_p/2\kappa_e$, where $C_p = 1.98 \times 10^3 \text{ J m}^{-3} \text{ K}^{-1}$ and $\kappa_e = 3.10 \text{ W m}^{-1} \text{ K}^{-1}$ are electronic specific heat and thermal conductivity of gold, respectively. Taking $L = 260$ nm as the half of the distance between the edges of two adjacent holes, we get an estimate $\tau_d \sim 2.2$ ps. As the diffusion time $\tau_d$ and carrier cooling time $\tau_p$ are comparable, an increase of 1 ps in $\tau_p$ can be possibly attributed to the diffusion of hot carriers. In addition, as the plasmon-induced hot carriers in graphene are generated through the hole edges (or plasmonic hotspots) in contact with graphene, hot carriers generated are initially nonuniformly distributed in graphene. The hot carriers in graphene also take finite time to diffuse and form a uniform distribution between the plasmonic hotspots. The diffusion coefficients of hot carriers in CVD graphene is $D_L = 5.5 \times 10^7 \text{ cm}^2 \text{s}^{-1}$. The characteristic diffusion time of hot carriers estimated by $\tau_d = L^2/2D_L$ gives $\tau_d = 57$ fs for $L = 250$ nm (half of the average distance between hole edges or plasmonic hotspots). The fast diffusion ensures uniform distribution hot carriers in graphene within the pump pulse duration. Therefore, the presence of graphene has no effect on $\tau_p$ in G-AuHA. However, a slightly reduced value of $\tau_p$ is observed in G-AuH in comparison to the Au film. As graphene is in more contact with the Au film, it modifies the decay rate to the effective decay rate ($1/\tau_{eff} = 1/\tau_p + 1/\tau_d$) and therefore results in a slightly faster decay.

A comparison of ultrafast carrier dynamics also helps us in identifying the direct or indirect nature of the plasmon-induced hot carrier generation in graphene. If the transfer pathway of hot carriers in graphene is indirect (due to the near-field interaction of graphene with gold) or plasmon-induced hot electron transfer (PHET), the carrier dynamics in graphene must be affected by the diffusion dynamics of hot carriers in AuHA and should show a slower decay dynamics. However, as the time constant $\tau_0$ observed by us in G-AuHA agrees with that reported for CVD graphene on the quartz substrate, we conclude that the transfer pathway of plasmon-induced hot carriers in graphene in the G-AuHA system is direct. The back transfer of hot carriers to Au is prevented by spatially separating them from the plasmonic hot spots (protruding edges of the holes).

IV. CONCLUSION

In conclusion, we have studied the photoexcited carrier dynamics of the hybrid plasmonic structure of the graphene-covered gold hole array to study the graphene-plasmon interaction. Graphene on the 3D hole array of gold shows strong coupling to the plasmonic modes. We have shown that plasmon-induced hot carriers can be
efficiently generated in graphene. The frequency tuning of EOT resonances in the Au hole array (by changing the hole spacing) is a very attractive possibility to inject hot carriers in graphene in spectral regions other than the plasmon resonances in noble metals in the visible region. Our results have interesting technological implications in hot carrier devices with tunable photon energy. The hot electrons once separated from the plasmonic hot spots can be utilized in surface catalytic reactions and optoelectronic devices. Similar to the reported photocatalysis using visible light overlapping with the surface plasmon resonance of the Au nanoparticles covered with graphene oxide,” we propose that a photocatalysis should be possible by graphene covered AuHA using photons of energy close to other plasmonic resonances of the AuHA (0.8–1.6 eV).

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