# Low-temperature synthesis and characterization of $Ag_2S_{1-x}Te_x(0 \le x \le 1)$

#### H N VASAN and A K SHUKLA

Solid State and Structural Chemistry Unit, Indian Institute of Science, Bangalore 560 012, India

MS received 15 January 1992

Abstract. A low-temperature route for the synthesis of  $Ag_2S$ ,  $Ag_2T$  and their solid solutions  $Ag_2S_{1-x}Te_x$  ( $0 \le x \le 1$ ) is reported.  $Ag_2S$  is prepared by the direct addition of silver nitrate solution to thiourea, while  $Ag_2T$  is prepared by reacting silver nitrate solution with tellurium in nitric acid and subsequently reducing it with hydrazine hydrate. The solid solutions of  $Ag_2S$  and  $Ag_2Te$  are obtained by the addition of nitrate solutions of silver and tellurium to thiourea followed by its reduction with hydrazine hydrate. The method enables the synthesis of low-temperature crystalline phase of  $Ag_2S_{1-x}Te_x$  solid solutions. The powder X-ray diffraction studies suggest that the solid solutions of compositions x < 0.3 have a phase akin to  $\alpha$ - $Ag_2S$  and those with compositions x > 0.6 are similar to  $\alpha$ - $Ag_2Te$ . In the intermediate range of compositions (x = 0.4 and 0.5), the solid solutions are found to be mixtures of  $\alpha$ - $Ag_2S$  and  $\alpha$ - $Ag_2Te$  phases which transform totally to  $\alpha$ - $Ag_2S$  phase on prolonged annealing at about 473 K.

Keywords. Low-temperature synthesis; silver chalcogenide; Ag<sub>2</sub>X characterization.

#### 1. Introduction

Silver chalcogenides,  $Ag_2X(X = S, Se, Te)$  being mixed conductors are potential materials for certain applications such as solid-state coulometers and battery electrodes. These materials undergo a structural  $\alpha$  to  $\beta$  phase transition at about 423 K accompanied by a marked increase in their electronic and ionic conductivities (Wagner 1953; Miyatani 1958, 1959, 1960; Shukla and Schmalzried 1979; Shukla et al 1981; Sohege and Funke 1984, 1989). One could synthesize these chalcogenides by the solid-state reaction of the constituent elements. But a similar method for preparing the solid solutions of Ag<sub>2</sub>S and Ag<sub>2</sub>Te results either in the formation of an amorphous phase or the high temperature  $\beta$ -phase which is retained even at room temperature (Miyatani 1960). In this study, we report a low-temperature route to synthesize phases of Ag<sub>2</sub>S, Ag<sub>2</sub>Te and Ag<sub>2</sub>S<sub>1-x</sub>Te<sub>x</sub> (0·1  $\leq x \leq$  0·9) similar to the method (Kulifay 1962) for synthesizing transition metal tellurides. These materials have been characterized by elemental analysis, powder X-ray diffraction (PXD) and differential scanning calorimetry (DSC). The room temperature PXD patterns of these solid solutions resemble α-phases of either Ag<sub>2</sub>S or Ag<sub>2</sub>Te depending on their composition. Unlike the parent compounds, which on heating undergo a distinct first order  $\alpha$ - $\beta$  transition, the solid solutions show a similar but broad phase transition.

### 2. Synthesis and characterization

Silver sulphide was prepared by mixing the solutions of  $AgNO_3(2 \times 10^2 \text{mol dm}^{-3})$  and  $CS(NH_2)_2(0.1 \text{ mol dm}^{-3})$  in water and heating around 353 K with constant

stirring for 2-3 h. The resulting black precipitate of Ag<sub>2</sub>S was filtered, washed with water followed by methanol and dried in air oven at about 373 K.

Silver telluride was prepared by dissolving tellurium powder (0.01 mol) in hot 1:1 nitric acid and water. The excess acid was neutralized with the desired quantity of dilute ammonia solution. This solution was then added to silver nitrate (0.2 mol dm<sup>-3</sup>) solution. The resulting solution was slowly added with constant stirring to 500 ml of 10% hydrazine hydrate in water having a pH of 9 which was raised to 10 by adding dilute ammonia solution and heating around 353 K with constant stirring for about 8 h. The black coloured  $Ag_2$ Te formed was washed and dried as before.

Thiourea hydrolyses in water giving  $H_2S$  (Pass and Sutcliffe 1968) and Te in tellurous acid was reduced from +4 to -2 by the addition of hydrazine hydrate. The plausible reaction route for the formation of  $Ag_2S$  and  $Ag_2Te$  may then be written as,

$$S = C(NH_2)_2 + 2H_2O \rightarrow 2NH_3 + H_2S + CO_2$$

$$2AgNO_3 + H_2S \rightarrow Ag_2S + 2HNO_3$$

$$2Ag^+ + Te^{+4} + N_2H_4 + OH^- \rightarrow Ag_2Te + N_2O + 5H_2O.$$

A similar procedure was adopted for the preparation of solid solutions  $Ag_2S_{1-x}Te_x$   $(0.1 \le x \le 0.9)$ . Solutions of thiourea, tellurous acid and silver nitrate were added in the molar ratio of 1-x:x:2 to 500 ml of 10% hydrazine hydrate solution in water at pH equal to 10. A portion of these compounds were annealed around 473 K in evacuated  $(10^{-5}$  torr) sealed tubes. Both the unannealed and annealed samples were characterized by PXD and DSC. The room temperature PXD patterns were recorded on a JEOL JDX-8P/70 X-ray diffractometer with  $CuK_{\alpha}(\lambda = 1.5418 \text{ Å})$  radiation. While the high temperature PXD patterns were obtained using a locally fabricated high temperature cell attached to PW 1050/70 Phillips X-ray diffractometer. The phase transformation temperature of the solid solutions were recorded on a Perkin Elmer DSC-2 differential scanning calorimeter.

## 3. Results and discussion

The elemental analysis showed a slight excess of Ag in all the solid solutions and hence only the nominal compositions are reported. The indexed room temperature PXD patterns of unannealed Ag<sub>2</sub>S, Ag<sub>2</sub>Te and two representative solid solutions Ag<sub>2</sub>S<sub>1-x</sub>Te<sub>x</sub> (x = 0.2 and 0.8) are shown in figures 1 and 2 and their indexed patterns with relative intensities are given in tables 1 and 2, respectively. The PXD patterns of Ag<sub>2</sub>S and Ag<sub>2</sub>Te match with the respective monoclinic  $\alpha$ -phases, whereas their solid solutions with compositions x < 0.3 are similar to  $\alpha$ -Ag<sub>2</sub>S phase and those of compositions x > 0.6 are akin to  $\alpha$ -Ag<sub>2</sub>Te phase. In the intermediate range (x = 0.4, 0.5), the products are mixtures of  $\alpha$ -Ag<sub>2</sub>S and  $\alpha$ -Ag<sub>2</sub>Te phases.

From tables 3 and 4, it is seen that within a system the lattice parameters of the solid solutions do not vary much in relation to their parent compounds although the atomic sizes of S(1.27 Å) and Te(1.60 Å) do differ. However, the solid solutions have different phase transition temperatures as seen from figure 3. The phase transformation temperatures of the parent compounds are in agreement with the values reported in the literature (Miyatani 1959). The phase transition temperature of the solid solutions continuously varies on the tellurium and sulphur-rich sides.

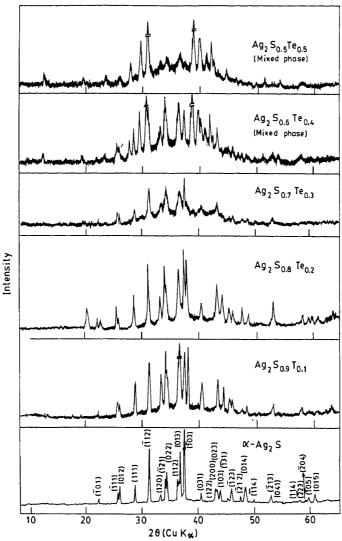


Figure 1. X-ray powder diffraction patterns of unannealed  $\alpha$ -Ag<sub>2</sub>S and solid solutions Ag<sub>2</sub>S<sub>1-x</sub>Te<sub>x</sub>( $x \le 0.5$ ).

Furthermore, all these phase transitions are reversible but are not as sharp as the parent compounds. It is interesting to note that the solid solution with x = 0.4 shows only one phase transition, while the composition x = 0.5 gives almost negligible enthalpy of phase transformation though mixtures of two phases are seen to be present in both the cases in their powder X-ray diffractograms.

Figure 4 shows the high temperature (523 K) PXD patterns of  $Ag_2S$  and  $Ag_2Te$  along with the four representative solid solutions. Of these the  $Ag_2S$  and  $Ag_2Te$  diffractograms resemble the bcc and fcc phases as reported in the powder diffraction files of inorganic compounds (4-0774 and 6-0575), while all the solid solutions but for x = 0.2 show an amorphous diffraction pattern. However, the solid solutions on cooling show poor crystallinity.

Table 1. Indexed X-ray lines of  $Ag_2S$  and  $Ag_2S_{0.8}Te_{0.2}$  (refined by least square method).

$Ag_2S$				$Ag_2S_{0\cdot8}Te_{0\cdot2}$			
hkl	$d_{ m obs}$	$d_{\mathrm{cal}}$	$I_{\mathrm{obs}}$	$d_{ m obs}$	$d_{\mathrm{cal}}$	$I_{\mathrm{obs}}$	
				4.291		26	
T01	3.952	3.953	4	3.952	3.960	16	
7	2.442	2 42 4	40	3.850		14	
Ī11	3.440	3.434	10	3.434	2 200	32	
012	3.389	3-388	17	3.376	3.378	14	
111	3.079	3.080	17	3.079	3.078	46	
112	2.840	2.837	53	2.886	2.837	88	
	2.683	2.644	2				
120	2.667	2.664	6	2.663	2.664	35	
_	2.644		2				
<u>1</u> 21	2.607	2.606	22	2.607	2.607	79	
022	2.582	2.586	29	2.585	2.581	50	
121				2.442	2.440	69	
112	2.458		22				
013	2.426	2.425	48	2.426	2.417	48	
103	2.383	2.384	100	2.385	2.380	100	
	2.257		3				
031	2.214	2.215	8	2.214	2.214	23	
122	2.094	2.095	4				
200	2.085	2.081	13	2.085	2.085	51	
023	2.076	2.074	12				
103	2.051	2.051	10	2.045	2.045	37	
Ĩ31	1.991	1.995	4	1.994	1.995	16	
<u>1</u> 23	1.965	1.964	12	1.967	1.962	16	
212	1.903	1.901	6	1.903	1.903	24	
014	1.870	1.870	14	1.868	1.863	14	
<u>1</u> 14	1.817	1.818	3	2 000	1 005	4.7	
213	1.719	1.717	8	1.717	1.718	33	
041	1.692	1.692	2		1 / 10	55	
114	1.610	1-612	1				
<u>1</u> 41	1.585	1-587	1				
223	1.579	1-578	3	1.580	1.579	14	
204	1.552	1.550	2	1.555	1.555	12	
<b>1</b> 05	1.542	1.542	6	1.541	1.538	10	
015	1.517	1.516	8	1.514	1.510	14	
Ī34			-				
'				1.459	1.458	13	
						22	

The PXD patterns and the phase transition temperatures of the annealed samples of the parent compounds and their solid solutions of compositions x = 0.1 and 0.8 are the same as of the unannealed sample. For the intermediate compositions (x = 0.4 - 0.6), diffused patterns were observed similar to the one reported by Koji and Iida (1985) for the samples prepared by conventional solid-state reactions. For the compositions x = 0.2, 0.3 and 0.7, the PXD patterns are poorly crystalline with a broad phase transition as shown in figure 5.

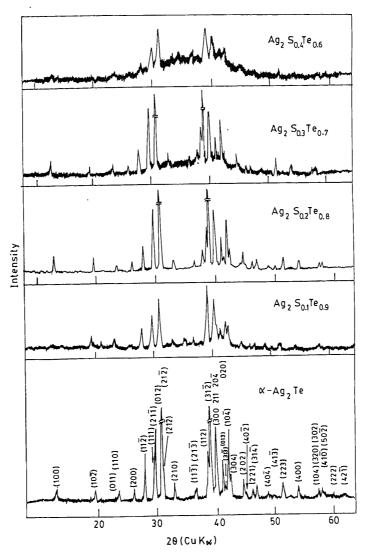


Figure 2. X-ray powder diffraction patterns of unannealed  $\alpha$ -Ag<sub>2</sub>Te and solid solutions Ag<sub>2</sub>S<sub>1-x</sub>Te<sub>x</sub>(x  $\geqslant$  0·6).

Table 2. Indexed X-ray lines of  $Ag_2Te$  and  $Ag_2S_{0\cdot 2}Te_{0\cdot 8}$ .

	$Ag_2$	Те	$Ag_2S_{0\cdot 2}Te_{0\cdot 8}$			
hkl	$d_{\mathrm{obs}}$	$d_{\mathrm{cal}}$	$I_{\mathrm{obs}}$	$d_{\mathrm{obs}}$	$d_{\mathrm{cal}}$	$I_{\mathrm{obs}}$
100	6.810	6.779	11	6.733	6.760	13
$10\overline{2}$	4.495	4.489	8	4.484	4.484	10
111	3.834	3.840	3		3.836	
002		_	_	3.754	3.758	5
110	3.739	3-742	7	3.723	3.734	6

(Continued)

Table 2. (Continued)

Ag <sub>2</sub> Te				$Ag_2S_{0\cdot 2}Te_{0\cdot 8}$			
hkl	$d_{ m obs}$	$d_{\rm cal}$	$I_{\mathrm{obs}}$	$d_{ m obs}$	$d_{\rm cal}$	$I_{\mathrm{obs}}$	
200	3-389	3.389	10	3.700	3.380	9	
112	3.175	3.174	29	3-159	3.168	24	
111	3.013	3.007	40		3.000		
211	2.993	2.995	62	2.979	2.986	57	
012	2.880	2.881	100		2.879	_	
212	2.858	2.870	30	2.867	2.858	100	
210	2.698	2.706	14	2.691	2.698	9	
113	2.452	2.452	9	2.447	2.449	6	
213	2.445	_		_	2.440	v	
_			_	2.359	-	16	
112	2.324	2.324	44	2-315	2.322	35	
312	2.308	2.309	100	2.295	2.301	85	
300		2.261	_		2.253	92	
211	2.254	2.251	64		2.248		
204		2.245		2.242	2.240	54	
020		2.240				54	
013	2.189	2.188	36	2.106	2.240		
313	2.171	2.174	16	2.186	2.187	28	
021		2.147	10	2.166	2.167	8	
121	2.145	2.145	60		2·147 2·145		
104		2:136	_	2.127			
304	2.125	2.125	23	2.137	2.135	45	
				2·117 2·041	2.118	16	
202	2.025	2.026	16	2.026	2024	4	
Ю <u>2</u>	2.010	2.007	3	2.006	2·024 2·003	16 4	
21	1.961	1.960	6	1.957			
14	1.935	1.929	10	1.95/	1.957	7	
14		1.920	10	1.007	1.927		
04	1.864	1.864	5	1.927	1.915	10	
_	_		<i></i>	1.855	1.856	4	
13	1.850	1.0/1	-	1.852		1	
13	1.826	1.841	3	1.841	1.840	1	
23	1.779	1·827 1·779	4	1.817	1.820	3	
2 <u>3</u>		1.777	14		1-778		
14	1.738	1.733	3	1.771	1.775	11	
_	1 /30	1.733	3	1.734	1.173	8	
- 2Ī		1.604	_	1.728	•	1	
00	1.695	1.694		1.692	1.691	8	
)4	1.604	1.695	10		1.690		
10	1.587	1·600 1·586	5	1.600	1.600	4	
)2	1 307		8	1.585	1.582	4	
12 2 <b>4</b>		1.581		1.575	1.575	1	
: <del>4</del> :4		1.546			1.545		
4	1.510	1.542		1.538	1.540	1	
2	1.512	1.507	2	1.506	1.507	1	
~	1-502	1.503	3	1.499	1.502	1	

(Continued)

Table 2. (Continued)

Ag <sub>2</sub> Te				$Ag_2S_{0\cdot 2}Te_{0\cdot 8}$			
hkl	$d_{ m obs}$	$d_{\mathrm{cal}}$	$I_{\mathrm{obs}}$	$d_{\mathrm{obs}}$	$d_{\mathrm{cal}}$	$I_{ m obs}$	
422		1.497		_	1.493		
306		1.497		_	1.493		
514		1.495	_		1.488		
321	1.448	1.448	15	1.443	1.446	13	
42 <u>1</u>		1.445	_		1.441		
			_	1.408		1	
223		1.395	_	1.397	1.394	12	
032		1.388		1.391	1.389	10	

**Table 3.** Lattice parameters of  $Ag_2S$  and solid solutions  $Ag_2S_{(1-x)}Te_x$  ( $x \le 0.3$ ).

Compound	a (Å)	<i>b</i> (Å)	(Å)	β	Cell vol (ų)	Density (d/g cm <sup>-3</sup> )
Ag <sub>2</sub> S	4.222(9)	6.93(2)	7.88(0)	99.58(8)	227-34	7-239
$Ag_2S_{0.9}Te_{0.1}$	4.23(3)	6.93(6)	7.92(7)	100.1(4)	228-57	7.478
$Ag_2S_{0.8}Te_{0.2}$	4.23(2)	6.93(4)	7.85(4)	99.7(1)	226.82	7.815
$Ag_2S_{0\cdot7}Te_{0\cdot3}$	4.23(3)	6.94(4)	7.86(5)	99.9(2)	227-37	8.075

**Table 4.** Lattice parameters of  $Ag_2Te$  and solid solutions  $Ag_2S_{(1-x)}Te_x(x \ge 0.6)$ .

Compound	a (Å)	<i>b</i> (Å)	c (Å)	β	Cell vol (ų)	Density (g cm <sup>-3</sup> )
Ag <sub>2</sub> Te	8·10(2)	4.488(9)	8-98(2)	123-15(7)	273-32	8-343
$Ag_2 Te_{0.9} S_{0.1}$	0.07(3)	4.47(2)	8.94(3)	123.0(7)	270.46	8-196
$Ag_2Te_{0.8}S_{0.2}$	8.10(2)	4.488(9)	8.98(2)	123.0(1)	273.78	7.875
$Ag_2Te_{0.7}S_{0.3}$	8.08(2)	4.47(2)	8-97(3)	123.04(9)	270.98	7.646
$Ag_2Te_{0.6}S_{0.4}$	8.08(2)	4.47(1)	8.97(3)	123.25(7)	270.94	7.414

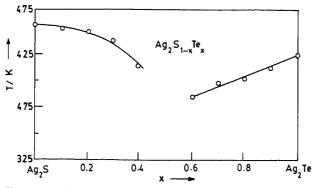


Figure 3. Phase transition temperature of  $Ag_2S$  and  $Ag_2Te$  and their solid solutions  $Ag_2S_{1-x}Te_x$  as a function of x.

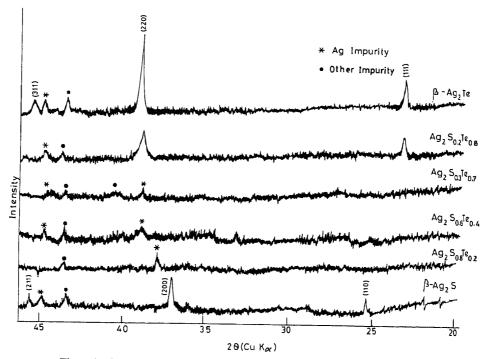


Figure 4. X-ray powder diffraction patterns of  $\beta$ -Ag<sub>2</sub>S,  $\beta$ -Ag<sub>2</sub>Te and Ag<sub>2</sub>S<sub>1-x</sub>Te<sub>x</sub> at 523K.

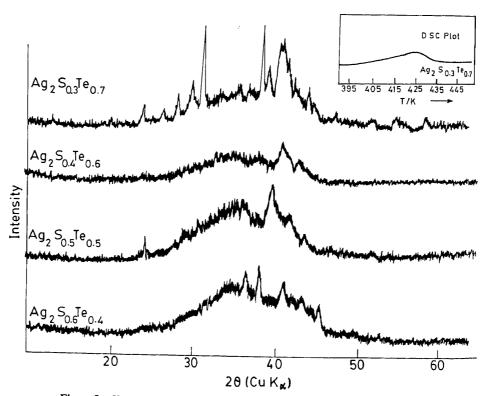


Figure 5. X-ray powder diffraction patterns of annealed  $Ag_2S_{1-x}Te_x$  (x=0.4-0.7); the inset shows a typical DSC curve for the solid solution of x=0.7.

## References

Koji H and Iida K 1985 J. Phys. Soc. Jpn 54 2218
Kulifay M S 1962 J. Am. Chem. Soc. 83 4961
Miyatani S 1958 J. Phys. Soc. Jpn 13 341
Miyatani S 1959 J. Phys. Soc. Jpn 14 1634
Miyatani S 1960 J. Phys. Soc. Jpn 15 1586
Pass G and Sutcliffe H 1968 Practical inorganic chemistry (London: Chapman and Hall)
Shukla A K and Schmalzried H 1979 Z. Phys. Chem. 59 118
Shukla A K, Vasan H N and Rao C N R 1981 Proc. R. Soc. (London) A376 619
Sohege J and Funke K 1984 Ber. Bunsenges. Phys. Chem. 88 657
Sohege J and Funke K 1989 Ber. Bunsenges. Phys. Chem. 93 115
Wagner C 1953 J. Chem. Phys. 21 1819