Microwave synthesis of electrode materials for lithium batteries

M HARISH BHAT, B P CHAKRAVARTHY, P A RAMAKRISHNAN, A LEVASSEUR † and K J RAO*

Solid State and Structural Chemistry Unit, Indian Institute of Science, Bangalore 560 012, India †ENSCPB, Avenue Pey Berland, BP 108, 334 02, Talence Cedex, Bordeaux, France

MS received 7 November 2000

Abstract. A novel microwave method is described for the preparation of electrode materials required for lithium batteries. The method is simple, fast and carried out in most cases with the same starting material as in conventional methods. Good crystallinity has been noted and lower temperatures of reaction has been inferred in cases where low temperature products have been identified.

Keywords. Novel microwave method; electrode materials; lithium batteries.

1. Introduction

There has been an increasing interest in the development of Li batteries for the last two decades, owing primarily to the higher voltages and energy densities realizable in them. The problems of safety associated with the use of Li metal due to its chemical reactivity with the ambient and the problems of dendritic regrowth of Li on the anode upon cycling, which leads to short-circuiting of the cell, have been the major disadvantages of Li batteries. This is sought to be overcome by the introduction of a more advanced and inherently safe 'rocking chair' configuration of lithium batteries (so called because the Li ions rock back and forth between the two electrodes during charge/discharge cycles). Rocking chair batteries use insertion compounds as electrode materials for both the anode and the cathode (Murphy et al 1978; Murphy and Carides 1979; Lazzari and Scrosati 1980; Mizushima et al 1980). The output voltage of such cells is determined by the difference between the electrochemical potential of Li⁺ ions in the two electrodic insertion compounds. It is implied that the cathode and anode should be compounds which can be intercalated with Li+ ions at high and low voltages respectively.

Intensive research on electrode materials has shown that for the negative electrode (anode), Li metal can be replaced effectively either by a Li–Al alloy (Peled *et al* 1983) or by intercalation materials based on carbon (Kanno *et al* 1989; Mohri *et al* 1989; Fong *et al* 1990). The loss in the voltage due to the Li⁺ potential in intercalated carbon compared to Li metal itself is compensated by the use of cathode materials with higher oxidation potentials. Some of the materials which meet this require-

Use of microwaves for the synthesis of inorganic compounds has gained great importance in recent years (Mingos and Baghurst 1991; Rao and Ramesh 1995; Rao et al 1999). The method offers several advantages over the conventional routes, the most important of them being the very short reaction times and energy economy. Several microwave reactions are now known to occur at lower temperatures than in conventional methods. The rapidity of the reactions offers excellent condition for retention of metastable phases. This novel method has been found to result in better reaction yields and better structural uniformity of products (Rao et al 1999a) than conventional ceramic methods. Further, the microwave method is also very clean and non-polluting. The rapidity of the reactions offers excellent possibilities for stabilizing metastable phases. This novel method has been found to result in better reaction yields and better structural uniformity of products. In order to use the microwave route, it is desirable that at least one of the reactants should be a microwave susceptor. However, this limitation can be overcome by the use of a secondary susceptor, which is chemically inert with respect to the reactants of interest. A micro-

ment are LiMn₂O₄ (Tarascon and Guyomard 1991, 1993; Guyomard and Tarascon 1992), LiCoO₂ (Thomas *et al* 1985; Gummow *et al* 1992) and LiNiO₂ (Dahn *et al* 1990, 1991; Ozawa 1994). In batteries employing glassy electrolytes, compounds like TiS₂, Cu₄O(PO₄)₂, CuBi₂O₄, Bi₄B₂O₉ have been used as cathodes (Jones and Akridge 1995). It has been found that in all these cases, the performance of the electrodes is significantly influenced by the method of their preparation. For example, samples synthesized at low temperatures show better cyclability but lower capacity than those prepared at higher temperatures. It is, therefore, necessary to develop methods of synthesis, which give rise to materials of better electrode characteristics and are also cheap and reproducible.

^{*}Author for correspondence

wave susceptor like graphite, which is generally nonreactive, is a superb secondary heater (Vaidhyanathan et al 1997). Researches at our laboratory and elsewhere have demonstrated the use of microwaves in the synthesis of a wide variety of oxides (Vaidhyanathan et al 1997; Baghurst and Mingos 1988; Baghurst et al 1988; Kladnig and Horn 1990), chalcogenides (Whittaker and Mingos 1992; Landry and Barron 1993; Kniep 1993; Vaidhyanathan et al 1995), carbides (Clark et al 1991; Ahmad et al 1991; Ramesh et al 1994), nitrides (Kiggans et al 1991; Ramesh and Rao 1995; Vaidhyanathan and Rao 1997a), silicides (Vaidhyanathan and Rao 1997b), borides (Mingos and Baghurst 1992) and glasses (Vaidhyanathan et al 1994, 1998). Therefore, we consider it important to develop a microwave route to synthesize electrode materials and examine their electrode characteristics. Materials like LiMn₂O₄, LiCoO₂, CuBi₂O₄ and Bi₄B₂O₉ have potential application in Li batteries. We report here a microwave method for the synthesis of these materials.

2. Experimental

Oxides or carbonates and iodides were used as reactants for the preparation of the complex oxides. Amorphous carbon was used as a secondary heater in some cases. Stoichiometric amounts of the required high purity reactants (total weight = 5-10 g) were ground well in each case and exposed to microwave irradiation for about 5-10 min in silica crucibles kept inside a domestic microwave oven (Batliboy-Eddy, 2.45 GHz, 980 W) operating at the highest power level. In some cases, the reaction was interrupted at the end of every 2 min and the partially reacted mixtures were ground well again to ensure homogeneity. Product formation was confirmed by powder X-ray diffraction (Philips, Model PW1050/70, Cu-K_a radiation). The lattice parameters were calculated using least squares fitting of XRD reflections.

Preliminary electrochemical characterization was carried out for LiMn₂O₄ by configuring appropriate cell assembly. Electrodes were made by mixing the active material

with PVDF (binder) and acetylene black in the ratio 80:10:10 by weight and pressing them as pellets. The pellets were heated to 175° C and were pressed once again at that temperature. A liquid electrolyte (LiClO₄ + EC + DEC) was used and was separated using polypropylene.

3. Results and discussion

$3.1 \quad LiMn_2O_4$

The microwave preparation of lithium manganate was first attempted with Li₂CO₃ and MnO₂ as reactants. But the expected cubic spinel phase of LiMn₂O₄ was not formed. Instead a mixed phase of defect spinels of Li₂O-yMnO₂ was found to form. This in itself is quite interesting because the defect spinel is reported to be stable only below 400°C and upon heating above 400°C, it transforms to the cubic spinel. Therefore, the metastable phase seems to get stabilized easily in microwave preparation. This may be due to low microwave susceptibility of MnO₂, which does not enable the system to attain high (> 400°C) temperature. The temperatures could not be measured in these experiments.

An alternate reaction scheme was used for the microwave preparation of the cubic spinel phase. An intimate mixture of LiI and MnO₂ was irradiated by microwaves. It was found that cubic spinel was formed directly in a single heating step of 6 min. The other product was iodine, which was in the gaseous phase and it provided an antioxidative shroud around the reaction mixture

$$LiI + 2 \ MnO_2 \rightarrow LiMn_2O_4 + \ \tfrac{1}{2} \ I_2.$$

Lattice parameter, a, of the cubic spinel phase was calculated by least squares fitting and found to be 8.244 Å. This value agrees very well with that reported in the literature (a = 8.2476 Å) [JCPDS, File No. 35-782]. The XRD pattern is shown in figure 1. The crystallinity of the product appears to be poor. The low microwave susceptibility of MnO₂ appears to limit the temperature attained by the

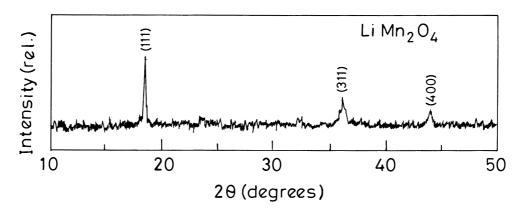


Figure 1. Powder X-ray diffraction pattern of microwave prepared cubic spinel, LiMn₂O₄.

reactants. Also the reaction does not appear to be highly exothermic. As a consequence the temperature attained appears insufficient for achieving good crystallinity.

Alternately, $LiMn_2O_4$ was prepared by using stoichiometric quantities of Li_2CO_3 and MnO_2 , but by embedding a pellet of the mixture in amorphous carbon (used as a secondary heater) before irradiating with microwaves. The reaction was found to be completed in about 7 min of continuous irradiation and a cubic spinel of $LiMn_2O_4$ was found to form:

$$\text{Li}_2\text{CO}_3 + 2\text{MnO}_2$$
 (pellet of mixture in carbon) \rightarrow $2\text{LiMn}_2\text{O}_4 + \text{CO}_2$.

However, the high temperature attained during the synthesis (since amorphous carbon is used as secondary heater) results in the formation of fine powders with somewhat poor crystallinity.

LiMn₂O₄ of spinel structure possesses cathode reversibility of intercalated Li⁺ ions above 4 V. Also, LiMn₂O₄ is not moisture sensitive and can be handled in ambient atmosphere. In comparison with the layered materials like LiCoO₂ and LiNiO₂, LiMn₂O₄ is cheaper and environmentally benign. Formation of spinel lithium manganate is very sensitive to preparative conditions. The conventional method of preparation of LiMn₂O₄ involved solid state reaction of Li₂CO₃ and Mn₂O₃ above 1150 K. Alternate synthetic routes based on precursor (Huang and Bruce 1994, 1995) and sol–gel (Barboux *et al* 1991) methods have also been reported. Microwave method described above is therefore, clearly faster, cleaner and economical than other methods.

A preliminary electrochemical characterization has been performed using two different anodes ($\text{Li}_{0.3}\text{V}_2\text{O}_5$ and graphitic carbon) having the configuration

$$\text{Li}_{0.3}\text{V}_2\text{O}_5 \mid \text{LiClO}_4 + \text{EC} + \text{DEC} \mid \text{LiMn}_2\text{O}_4,$$
 graphitic C $\mid \text{LiClO}_4 + \text{EC} + \text{DEC} \mid \text{LiMn}_2\text{O}_4.$

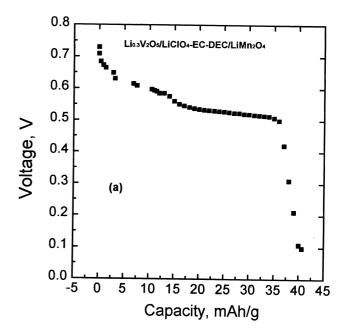
Batteries fabricated using Li_{0.3}V₂O₅ and LiMn₂O₄ as electrodes showed low voltages as well as low capacities (figure 2a). Note here that the Li_{0.3}V₂O₅ used was also prepared by a microwave method reported elsewhere (Rao *et al* 1999b). The low voltage of this battery is due to the small difference in the potentials of the electrode materials (w.r.t. Li). The expected voltages being $E \approx 3.4~V$ for Li_{0.3}V₂O₅/Li and $E \approx 4.2~V$ for LiMn₂O₄/Li.

Poor intercalatability of Li into the electrode materials results in low capacity. When graphitic carbon was used as an anode instead of $\text{Li}_{0.3}\text{V}_2\text{O}_5$ in the above cell, the voltage was about 3.8 V but the capacity was not very high (figure 2b). Further studies are in progress.

$3.2 \quad LiCoO_2$

The layered lithium cobaltate is an important cathode material and has been employed in the commercial batteries

manufactured by Sony Energytec (Nagaura 1990). This material was prepared from LiOH and Co(NO₃)₂·6H₂O by mixing them in distilled water in 4 : 1 molar ratio (LiOH taken in excess). The solution was microwave irradiated for 10 min in order to remove the water. Since neither of the reactant is a microwave susceptor, the temperature did not rise further. The product formed at this stage was amorphous. The sample was heated at 973°C for 60 min and X-ray pattern showed the formation of poorly crystalline LiCoO₂ (figure 3a). Alternately, Li₂CO₃ and Co(NO₃)₂·6H₂O were ground into a pellet and embedded



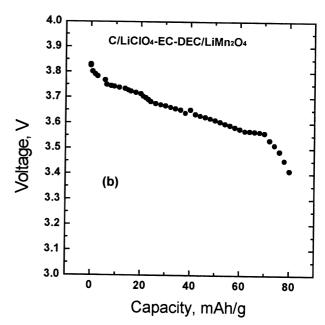
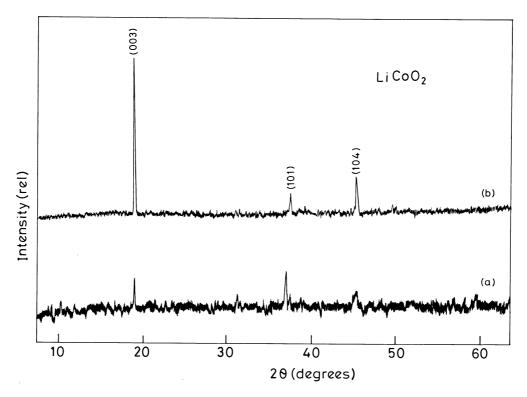


Figure 2. Discharge characteristics of LiMn₂O₄ using the anodes (a) Li_{0.3}V₂O₅ and (b) graphitic carbon, discharged at a constant current of 0.2 mA/cm².



 $\textbf{Figure 3.} \quad \text{Powder X-ray diffraction pattern of layered } LiCoO_2 \text{ prepared by (a) solution route and (b) solid state route.}$

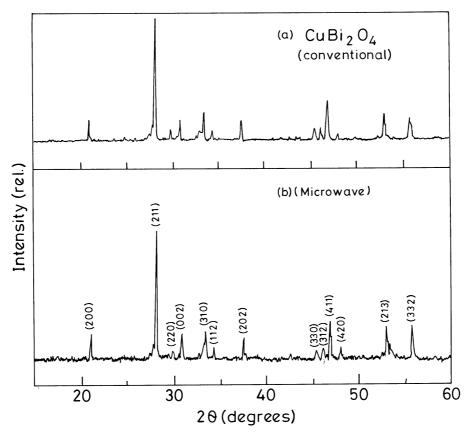


Figure 4. Powder X-ray diffraction pattern of $CuBi_2O_4$ prepared by (a) conventional and (b) microwave methods.

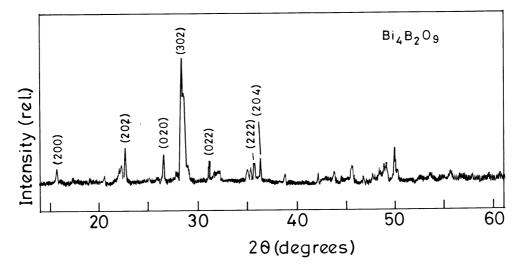


Figure 5. X-ray pattern of Bi₄B₂O₉ prepared by microwave route.

in amorphous carbon, which gave LiCoO₂ possessing very good crystallinity (figure 3b) within 5 min of irradiation in microwaves.

$3.3 \quad CuBi_2O_4$

Microwave irradiation of a mixture of CuO and Bi₂O₃ for just 5 min yielded monophasic, tetragonal CuBi₂O₄ (figure 4a) with lattice parameters a = 8.504 Å and c =5.811 Å. These lattice parameter values are in good agreement with literature reports (a = 8.510 Å and c = 5.814 Å) [JCPDS File no. 26-502]. Alternately, when a mixture of CuO and (BiO)₂CO₃ was pelletized and embedded in amorphous carbon, the reaction was complete in under a minute. For purposes of comparison, CuBi₂O₄ was also prepared using the conventional procedure (Boivin et al 1973). In this procedure, the initial reactants (CuO and Bi₂O₃) are heated at 523 K for 2 h following which two additional heat treatments are given at 623 and 723 K for 6 h each. At every stage, the powders are ground repeatedly. Final heat treatment is given at 873 K for 4 h. XRD (figure 4b) of the material obtained by this method gave lattice parameters a = 8.503 Å and c = 5.817 Å. The superiority of the microwave method, which gives the product in a single step irradiation and avoids the several steps involved in the conventional method, is self-evident. Also, the microwave method yields CuBi₂O₄ of better crystallinity as can be seen from the XRD patterns.

$3.4 \quad Bi_4B_2O_9$

The compound Bi₄B₂O₉ was synthesized by a microwave method by grinding (BiO)₂CO₃ and H₃BO₃ into a pellet, which was placed in amorphous carbon and irradiated

Table 1. Microwave prepared electrode materials and their preparative conditions.

No	Electrode . material	Reactants	Time (min)
1	LiMn ₂ O ₄	$\text{Li}_2\text{CO}_3 + \text{MnO}_2*$ $\text{LiI} + \text{MnO}_2$	5 6
2	CuBi ₂ O ₄	$\begin{array}{l} CuO + Bi_2O_3 \\ CuO + (BiO)_2CO_3* \end{array}$	5 1
3	LiCoO ₂	$\begin{array}{l} LiOH + Co \; (NO_3)_2 \cdot 6H_2O^{\#} \\ Li_2CO_3 + Co \; (NO_3)_2 \cdot 6H_2O^{*} \end{array}$	10 5
4	$Bi_4B_2O_9$	$(BiO)_2CO_3 + H_3BO_3$	1 (20% power) + 3 (100% power)

^{*}indicates that reactants were pelletized and placed in amorphous carbon (secondary heater);

with microwaves. The initial heating was done at low power (20%) for about a minute for the removal of water and subsequently it was kept at high power for 3 min. The product was confirmed to be monophasic $Bi_4B_2O_9$ using X-ray diffraction (figure 5). In comparison, the conventional method (Levin and McDaniel 1962) is tedious and involves several intermittent grindings.

The various electrode materials prepared by the microwave route and their preparative conditions are summarized in table 1.

4. Conclusions

Electrode materials, which find potential applications in Li batteries, have been synthesized by a microwave route for the first time. The method should prove to be very attractive for large-scale production in view of the very short time involved for their synthesis.

^{*}indicates solution route using microwaves.

Acknowledgement

Thanks are due to European Commission for the financial support through the project INCO-DC 950 400.

References

- Ahmad I, Dalton R C and Clark D E 1991 J. Microwave Power & Electromagnetic Energy 26 128
- Baghurst D R and Mingos D M P 1988 J. Chem. Soc., Chem. Commun. 829
- Baghurst D R, Chippindale A M and Mingos D M P 1988 Nature 332 311
- Barboux P, Tarascon J M and Shokoohi F K 1991 J. Solid State Chem. 94 185
- Boivin M J C, Thomas D and Tridot G 1973 Compt. Rend. C276 1105
- Clark D E, Ahmad I and Dalton R C 1991 Mater. Sci. Engg.
- Dahn J R, von Sacken U and Michal C A 1990 Solid State Ionics 44 87
- Dahn J R, von Sacken U, Juzkow M W and Janaby H A 1991 J. Electrochem. Soc. 138 2207
- Fong R, Von Sacken U and Dahn J R 1990 J. Electrochem. Soc. 137 2009
- Gummow R J, Thackery M M, David W I F and Hull S 1992 *Mater. Res. Bull.* **27** 327
- Guyomard D and Tarascon J M 1992 J. Electrochem. Soc. 139 937
- Huang H and Bruce P G 1994 *J. Electrochem. Soc.* **141** L76 Huang H and Bruce P G 1995 *J. Power Sources* **54** 52
- Jones S D and Akridge J R 1995 in *Handbook of solid state batteries and capacitors* (ed.) M Z A Munshi (Singapore: World Scientific) p. 209
- Kanno R, Takeda Y, Ichikawa T, Nakanashi K and Yamamoto O 1989 *J. Power Sources* **26** 535
- Kiggans J O, Hubbard C R, Steele R R, Kimrey H D, Holcombe C E and Tiegs T N 1991 *Ceram. Trans.* **21** 403
- Kladnig W F and Horn J E 1990 Ceram. Int. 16 99
- Kniep R 1993 Angew. Chem. 32 1411
- Landry C C and Barron A R 1993 Science 260 1653
- Lazzari M and Scrosati B 1980 J. Electrochem. Soc. 127 773

- Levin E M and McDaniel C L 1962 J. Am. Ceram. Soc. 45
- Mingos D M P and Baghurst D R 1991 Chem. Soc. Rev. 20 1
- Mingos D M P and Baghurst D R 1992 Br. Ceram. Trans. J. 91 124
- Mizushima K, Jones P C, Wiseman P J and Goodenough J B 1980 Mater. Res. Bull. 15 783
- Mohri M et al 1989 J. Power Sources 26 545
- Murphy D W and Carides J N 1979 J. Electrochem. Soc. 126 349
- Murphy D W, De Salvo F J, Carides J N and Waszczak J V 1978 Mater. Res. Bull. 13 1395
- Nagaura T 1990 Prog. Batt. Col. Cells 9 20
- Ozawa K 1994 Solid State Ionics 69 212
- Peled E, Lombardi A and Schlaikjer C R 1983 *J. Electrochem.* Soc. 130 1362
- Ramesh P D and Rao K J 1995 Adv. Mater. 7 177
- Ramesh P D, Vaidhyanathan B, Ganguli M and Rao K J 1994 J. Mater. Res. 9 3057
- Rao K J and Ramesh P D 1995 Bull. Mater. Sci. 18 447
- Rao K J, Vaidhyanathan B, Ganguli M and Ramakrishnan P A 1999a *Chem. Mater.* 11 882
- Rao K J, Ramakrishnan P A and Gadagkar R 1999b J. Solid State Chem. 148 100
- Tarascon J M and Guyomard D 1991 J. Electrochem. Soc. 138
- Tarascon J M and Guyomard D 1993 *Electrochim. Acta* 38 1221
- Thomas M G S R, Bruce P G and Goodenough J B 1985 J. Electrochem. Soc. 132 1521
- Vaidhyanathan B and Rao K J 1997a Chem. Mater. 9 1196
- Vaidhyanathan B and Rao K J 1997b J. Mater. Res. 12 3225
- Vaidhyanathan B, Ganguli M and Rao K J 1994 *J. Solid State Chem.* **113** 448
- Vaidhyanathan B, Ganguli M and Rao K J 1995 Mater. Res. Bull. 30 1173
- Vaidhyanathan B, Raizada P and Rao K J 1997 J. Mater. Sci. Lett. 16 2022
- Vaidhyanathan B, Prem Kumar C and Rao K J 1998 J. Phys. Chem. Solids 59 121
- Whittaker A G and Mingos D M P 1992 J. Chem. Soc., Dalton Trans. 2751